

# Effective core potential ab initio calculations on main group heptoxides and large silicate systems

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## Abstract

The ab initio molecular structures for several main group heptoxides ( $X_2O_7^{n-}$ ,  $n = 0, 2, 4$ ) are calculated using effective core potentials at the HF and DFT (B3LYP) levels. Particular attention is given to the X–O–X bond angle, as this structural parameter is a key feature for the study of both heptoxides and larger derivatives, such as polysilicate systems. The extent of the p–d interactions, which was found to be the main factor governing the magnitude of the X–O–X angle in transition metal heptoxides, does not play a significant role in the main group analogues.

In the cyclic polysilicate systems (up to 12 silicon atoms), the Si–O–Si bond angle is found to vary from 132° in small rings to 150° in the less strained systems. The results are in agreement with the experimental data available and its accuracy competes with that of results from much heavier all-electron calculations. © 2000 Elsevier Science B.V. All rights reserved.

*Keywords:* Heptoxides; Silicate; Zeolites; Molecular structures; Effective core potentials; B3LYP method

## 1. Introduction

In a recent publication [1] we have reported the calculated ab initio molecular structures and vibrational frequencies for the heptoxides of the VB, VIB and VIIB transition metal groups (general formula  $X_2O_7^{n-}$ , see Scheme 1 below), by using effective core potentials (ECP) at both HF and B3LYP levels. The relative energies of all optimized configurations, as well as the particular conformational preferences of each heptoxide, were presented and discussed. Special emphasis was given to the magnitude of the X–O–X bond angle, as it raised some attention in the last few years [1–13].

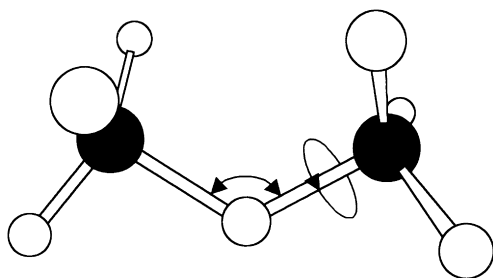
A general good agreement between the calculated

and the reported experimental values, for both structural parameters and vibrational frequencies, was found by using the therein-selected ECP combination [1]. The overall results showed a clear tendency to linearity of the X–O–X framework on going both downwards a group and backwards a period (VIIB → VB), with the VB heptoxides presenting a perfectly centrosymmetric molecular geometry. The systematic behaviour of the X–O–X bond angle arises from size-dependent steric interactions, magnitude of the total heptoxide charge and extent of the p–d interactions. Based on the calculated energy differences between the 2p orbitals of an  $O^{2-}$  anion and the valence d orbitals of the metal cation, it was concluded that this latter effect is particularly meaningful for the reported transition metal heptoxides [1].

The wide practical applications of polysilicate analogues, obtained by substitution of some silicon

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Scheme 1.

atoms by other type of elements of both main (Al, P, Ge, Sn, Te, among others) and transition metal (e.g. V and Ti) groups, have renewed the interest of the structure–property relationships of X–O–X containing systems. Several studies of large polysilicate analogues using *ab initio* methods at different levels of sophistication have recently been reported [14–18]. Such calculations are highly demanding in computational resources and its application as an additional laboratory tool requires the use of powerful computers. Thus, the evaluation of the accuracy and reliability of light ECP/basis set combinations is still of practical interest.

In this way, the purpose of the present study is twofold. Firstly, to determine the structural preferences of VA, VIA and VIIA main group heptoxides, following the previous report on the transition metal analogues [1]. Secondly, to evaluate the performance of the selected ECP-basis set combination for silicon containing systems, by comparing the optimized structures of some neutral cyclic polysilicon systems with reported experimental and/or higher level theoretical results available.

## 2. Computational methodology

The calculations were performed using the GAUSSIAN 98W program package [19] on a Pentium II 300 MHz computer with 288 MB of RAM.

All the calculations on  $X_2O_7^{n-}$  systems were performed by using the oxygen ECP/minimal valence basis set of Stevens et al. [20] deliberately augmented with a polarization d-type function ( $\zeta = 0.8$  [21]) which proved to be essential for the description of the X–O–X bond angle in this kind of system [1,3–

13]. It should be mentioned that the use of a double-zeta valence basis set for the oxygen atoms did not improve the quality of the results (see Ref. [1] and this work) and has been abandoned to keep the quality/computational costs ratio as high as possible. The hydrogen atom was described by the standard 3-21G basis set [22], while for the remaining elements but the halogen atoms, the Hay and Wadt's ECP [23] with double-zeta split valence shell was used. This ECP has proven elsewhere [24] to yield good results for main group organometallic compounds.

In the particular case of the VIIA group elements, a d-type polarization function was added to the Hay and Wadt's double-zeta valence space ( $\zeta = 0.75, 0.389$  and  $0.266$  for Cl, Br and I, respectively [21,25]) to overcome stability problems (e.g. in the absence of such polarization function the  $Cl_2O_7$  molecule splits into three fragments during geometry optimization, a problem that is not overcome by additional improvement of the oxygen atom basis set). Additional test calculations using dichloride monoxide ( $Cl_2O$ ) showed that the inclusion of a d-type polarization function on the chlorine atom leads to a good agreement between the calculated and experimental structural parameters of  $Cl_2O$  (168 pm and  $111^\circ$  vs. 170 pm and  $111^\circ$  [26], respectively).

In order to account for the electron correlation effects, the widely used hybrid method denoted by B3LYP [27–32], which includes a mixture of HF and DFT exchange terms and the gradient-corrected correlation functional of Lee et al. [33], as proposed and parameterized by Becke [34], was used (with standard B3LYP options of G98w).

Several possible configurations were used as starting points (see Fig. 1 of Ref. [1]) and fully optimized at the HF and B3LYP levels. The order of the critical points ( $\lambda = 0$  for minima,  $\lambda = 1, 2, \dots$  for first and higher order saddle points) was evaluated at the HF level by computing the vibrational frequencies.

For the silicon containing systems, an additional d-type function ( $\zeta = 0.39$ ) was included in the silicon atom basis set. The inclusion of this polarization function has been previously considered to be required on silicate derivatives [11–13], and it has been found to be the only additional requirement to obtain a good description of the neutral disilicate systems  $H_3Si-O-SiH_3$  and  $(OH)_3Si-O-Si(OH)_3$  with the present ECP/basis set combination,

Table 1

Calculated energy differences ( $\Delta E/\text{kJ mol}^{-1}$ ) between the several optimized geometries, relative to the energy minimum for each heptoxide studied (DFT = B3LYP). The number of calculated imaginary vibrational frequencies,  $\lambda$  (order of the critical point (0, 1 and 2 stand for minimum, first- and second-order saddle points, respectively)), are also included. As referred to in the experimental section, the valence shell basis set of the halogen atoms were augmented with a polarization function

Structure	$\text{P}_2\text{O}_7^{4-}$			$\text{S}_2\text{O}_7^{2-}$			$\text{Cl}_2\text{O}_7$			$\text{As}_2\text{O}_7^{4-}$			$\text{Se}_2\text{O}_7^{2-}$		
	HF	$\lambda$	DFT	HF	$\lambda$	DFT	HF	$\lambda$	DFT	HF	$\lambda$	DFT	HF	$\lambda$	DFT
$C_2$	0.00	0	0.00	0.00	0	0.00	0.00	0	0.00	0.00	0	0.00	0.00	0	0.00
$C_s$	0.09	1	0.20	1.43	1	0.89	4.45	1	3.19	0.05	1	0.17	0.73	1	0.58
$C_{2v(\text{syn})}$	4.54	1	1.69	7.52	1	1.57	3.62	1	0.75	4.43	1	1.89	8.30	1	2.88
$C_{2v(\text{anti})}$	6.94	2	4.24	23.63	2	10.27	28.12	2	17.11	6.06	2	4.09	17.80	2	10.34
$D_{3d}$	9.44	1	11.69	71.83	1	59.54	123.20	1	115.76	6.90	1	10.84	50.79	1	50.03
$D_{3h}$	12.84	2	13.51	74.28	2	60.63	123.68	1	116.25	10.25	2	12.71	53.50	2	51.34
	$\text{Br}_2\text{O}_7$			$\text{Sb}_2\text{O}_7^{4-}$			$\text{Te}_2\text{O}_7^{2-}$			$\text{I}_2\text{O}_7$			$\text{Bi}_2\text{O}_7^{4-}$		
	HF	$\lambda$	DFT	HF	$\lambda$	DFT	HF	$\lambda$	DFT	HF	$\lambda$	DFT	HF	$\lambda$	DFT
$C_2$	0.00	0	0.00	0.00	0	0.00	0.00	0	0.00	0.00	0	0.00	0.00	0	0.01
$C_s$	1.36	1	1.94	0.00	1	0.01	0.03	1	0.17	0.34	1	0.34	0.00	1	0.00
$C_{2v(\text{syn})}$	2.08	1	1.83	3.87	1	2.30	0.87	1	<sup>a</sup>	5.23	1	3.18	3.33	1	1.78
$C_{2v(\text{anti})}$	10.16	2	12.79	3.98	2	3.28	1.30	2	9.42	10.11	2	8.14	3.43	2	2.70
$D_{3d}$	31.15	1	47.57	0.67	1	4.95	1.81	1	29.40	28.75	1	36.08	0.53	1	5.15
$D_{3h}$	31.02	1	47.35	4.12	2	7.02	2.15	2	31.32	30.04	2	36.78	3.55	2	6.84

<sup>a</sup> Converged to  $C_2$  symmetry during geometry optimization.

particularly in what concerns the oxygen bridging bond angle.

### 3. Results and discussion

#### 3.1. $\text{X}_2\text{O}_7^{n-}$ systems

Table 1 shows the calculated energy differences between the several optimized geometries and the minimum energy configuration, found at the HF and B3LYP levels, for the several main group heptoxides herein considered. The number of imaginary frequencies ( $\lambda$ ) obtained at the HF level for each optimized structure is also included, showing that only one energy minimum (staggered  $C_2$  configuration) was found for each heptoxide. However, some caution should be taken on this subject, as the lowest wavenumbers for several minima are quite low ( $<10 \text{ cm}^{-1}$ ) and the HF energy difference between the lowest energy configurations is often smaller than a tenth of  $\text{kJ mol}^{-1}$ . In addition, the inclusion of the electron

correlation effects leads to a general narrowing of the energy differences between the stationary points.

The calculated structural parameters obtained for the minimum energy configuration of each heptoxide considered, at both HF and DFT levels, are presented in Table 2, in comparison with experimental data available. A general overestimation of the bond lengths is observed for the B3LYP geometries. This problem seems to be related with the B3LYP approximation, possibly resulting from the neglect of dispersion interactions.

The agreement between the calculated and the experimental values is good, if we keep in mind that the ab initio calculations consider the isolated molecules and the experimental data refer to solid samples, in which the heptoxide structure is conditioned by the crystal packing (except for  $\text{Cl}_2\text{O}_7$ ). The absence of the crystal environment is particularly critical for the multiply charged  $\text{X}_2\text{O}_7^{4-}$  anions, as it will be discussed below.

The best system to evaluate the quality of the calculated structural parameters is  $\text{Cl}_2\text{O}_7$ , since both

Table 2

Optimized parameters for the energy-minimum of each metal heptoxide studied. Experimental results are included, whenever existent (bond lengths are average values). Please note that, as referred to in the experimental section, the valence shell basis set of the halogen atoms were augmented with a d-type polarization function ( $O_b$  and  $O_t$  stand for bridge and terminal oxygen respectively; DFT = B3LYP)

$X_2O_7^{n-}$		X– $O_b$ (pm)	X– $O_t$ (pm)	$O_b$ –X– $O_t$ (°)	$O_t$ –X– $O_t$ (°)	X– $O_b$ –X (°)	Sym.
$P_2O_7^{4-}$	Calculated HF	172	156	103–110	112	148	$C_2$
	Calculated DFT	192	162	101–112	112	146	$C_2$
	Experimental <sup>a</sup>	161	152	101–110	109–114	133 143 <sup>a</sup>	
$As_2O_7^{4-}$	Calculated HF	180	167	103–110	112	149	$C_2$
	Calculated DFT	197	173	101–112	112	146	$C_2$
	Experimental <sup>b</sup>	179	168	102–107	111–118	116	
$Sb_2O_7^{4-}$	Calculated HF	194	184	105–109	113	161	$C_2$
	Calculated DFT	208	190	102–112	112	150	$C_2$
$Bi_2O_7^{4-}$	Calculated HF	203	193	105–110	112–113	162	$C_2$
	Calculated DFT	218	198	105–114	111	151	$C_2$
$S_2O_7^{2-}$	Calculated HF	170	152	100–108	114	126	$C_2$
	Calculated DFT	192	159	99–109	114	125	$C_2$
	Experimental <sup>c</sup>	165	144	101–106	113–116	124	
$Se_2O_7^{2-}$	Calculated HF	178	163	101–107	114	128	$C_2$
	Calculated DFT	196	169	99–109	114	124	$C_2$
$Te_2O_7^{2-}$	Calculated HF	188	177	102–106	114	136	$C_2$
	Calculated DFT	202	182	99–107	113–116	128	$C_2$
$Cl_2O_7$	Calculated HF	166	139	99–105	115	122	$C_2$
	HF/6-31G <sup>*d</sup>	168	140	99–105	115	122	$C_2$
	Calculated DFT	180	145	97–105	115–116	117	$C_2$
	MP2/6-31G <sup>*d</sup>	180	145	97–105	116	115	$C_2$
	Experimental <sup>e</sup>	170	140	98–106	115	119	$C_{2v}$
$Br_2O_7$	Calculated HF	173	144	101–103	116	129	$C_2$
	Calculated DFT	183	149	99–103	116	120	$C_2$
$I_2O_7$	Calculated HF	190	171	100–103	116	130	$C_2$
	Calculated DFT	200	176	99–103	116	124	$C_2$

<sup>a</sup> X-ray values for  $Hg_2P_2O_7$  [35]; value of 143° was reported previously for  $Co_2P_2O_7$  [36].

<sup>b</sup> X-ray values for  $Na_4Ce(As_2O_7)_7$  [37]; values of 127 and 128° are also encountered [38,39].

<sup>c</sup> X-ray values for  $K_2S_2O_7$  [40].

<sup>d</sup> All electron ab initio results [41].

<sup>e</sup> Gas-phase electron diffraction value [42].

gas-phase experimental results [42] and higher level ab initio calculations [41] have been reported for this molecule. As it can be seen from Table 2, there is an excellent agreement between the present results and the reported 6-31G<sup>\*</sup> values, thus supporting the quality of the ECP/basis set chosen. In particular, the bond lengths and bond angles obtained at the B3LYP level are nearly identical to those reported from much heavier MP2/6-31G<sup>\*</sup> calculations. The description of the experimental parameters is good, particularly that which concerns the bond angles. Interestingly, the experimental bond lengths of  $Cl_2O_7$  are better reproduced by the non-correlated methods

(even taken into account the different meanings of  $r_o$  and  $r_e$  structures).

The relative importance of the  $C_2$  and  $C_{2v}$  structures of  $Cl_2O_7$  deserves some attention. While X-ray diffraction results point to a  $C_{2v}$  geometry in the crystal [43], both calculations (this work and previous all-electron results [41]) and gas-phase electron diffraction results [42] agree in a  $C_2$  minimum for the isolated molecule. Since both structures ( $C_2$  and  $C_{2v}$ ) are calculated to be separated by only 0.75 kJ mol<sup>-1</sup> (at the B3LYP level), their relative energies can be easily reversed by crystal packing forces.

By analysing the calculated values for the X–O–X angle in Table 2, it is obvious that its variation follows the same general trends previously observed for the transition metal heptoxides [1]. In fact, the main group heptoxides also display a general tendency for linearization of the X–O–X framework on going both downwards a group and backwards a period (VIIA → VA). The comparison between the two sets of heptoxides (transition metal vs. main group) provides additional information concerning the factors governing the magnitude of the oxygen bridging angle. For the transition metal heptoxides [1], the observed trends were explained by three main effects: (i) total electronic charge of the heptoxide; (ii) metal atom size and (iii)  $\pi$ -bonding between the two filled 2p orbitals of the bridging oxygen atom and the empty d orbitals of the metal atom. The extent of the p–d interactions, which was concluded to be the dominant factor in the transition metal series [1], is not expected to play a significant role on the main group set. For instance, previous calculations on silicate derivatives show that a d-type function is required on the silicon atom, but only to polarize the s and p orbitals and give a better description of those orbitals at large distances from the nucleus [4,11,13].

The present results are in agreement with this observation. The increase of the X–O–X bond angle downwards a group is much less pronounced in the main group heptoxides than in the transition metal analogues series (ca. 5°, compared with more than 20°, respectively). Such a small variation can be explained by the increasing size of the X-atom along a group.

On the other hand, the main factor determining the increase of the bridging bond angle backwards each period can be evaluated by comparing the molecular structure of the fully charged VA anions with the structure of their protonated species  $H_mX_2O_7^{-4+m}$  ( $m = 1, \dots, 4$ ), whose net negative charge is progressively reduced. As expected, the lowest energy structures for the protonated species present intramolecular XOH...OX hydrogen bonds, which become the dominant factor controlling the X–O–X bond angle. However, considering only the optimized structures not presenting this type of interactions, there is a ca. 25° decrease of the X–O–X bond angle on going from the  $X_2O_7^{4-}$  to the  $H_2X_2O_7^{2-}$  species. Further

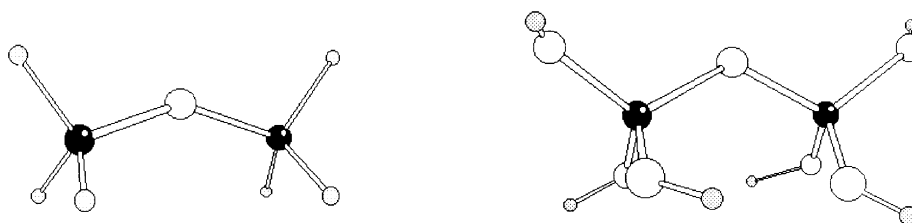
reduction of the net negative charge does not produce relevant changes, and the  $H_2X_2O_7^{2-}$ ,  $H_3X_2O_7^-$  and  $H_4X_2O_7$  species have nearly identical X–O–X angles, close to the values observed for the neutral heptoxides of group VIIA ( $Cl_2O_7$ ,  $Br_2O_7$  and  $I_2O_7$ ). Thus, the large net electronic charge is the only factor determining the wide X–O–X bond angles calculated for the isolated main group VA heptoxides. This contrasts with the case of the transition metal heptoxides, for which linear X–O–X angles were observed for the VB anions—with the same net charge of the group VA anions—because of the presence of additional d–p interactions [1].

### 3.2. Silicon-containing systems

As a first test for the performance of the selected ECP/basis set for silicon containing systems, the molecular structures of the neutral silicon analogues  $H_3Si-O-SiH_3$  and  $(OH)_3Si-O-Si(OH)_3$  were determined. Fig. 1 shows the optimized structures for both systems compared with the reported theoretical and experimental data.

Both forms present a clear preference for a bent Si–O–Si angle, in agreement with previous experimental and higher level ab initio calculations [12,13,44]. In the case of the -OH derivative, the occurrence of intramolecular hydrogen bond contacts in the isolated molecule leads to an Si–O–Si bond angle below 140°. It should be stressed that the current calculations yield an excellent description of the reported gas-phase diffraction structure of disiloxane [44], significantly better than the much heavier HF/6-31G\* calculations [13]. The better performance is notorious for the Si–O–Si bond angle, which is highly overestimated at the HF/6-31G\* level (Si–O–Si of 169° [13]) while it is perfectly described by the present HF calculations (and overestimated by just 3° at the B3LYP level). This is particularly important for the extension of the present calculations to larger silicon containing systems. In fact, due to the well-known flexibility of the Si–O–Si framework, the correct prediction of the Si–O–Si bond angle is a key feature for the ab initio structural study of silicon systems.

Figs. 2 and 3 present the calculated structures for the cyclic trimeric and tetrameric siloxane analogues with hydrogen and chloro terminal atoms, respectively,

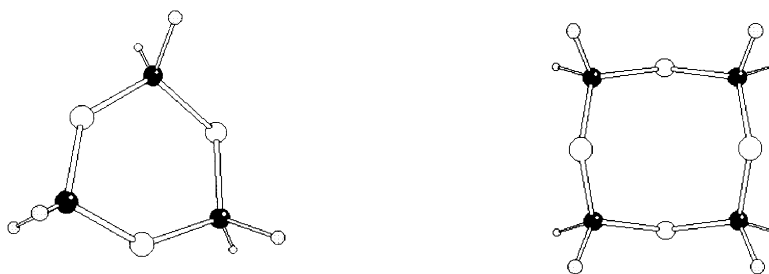


HF	B3LYP	Exp.	MP2/6-31G*		HF	B3LYP	Exp.	6-31G*
162	164	163	166	Si – O <sub>b</sub>	162	164	162	162
148	149	149	148	Si – •	162	163		162
110	110	110	110	O <sub>b</sub> – Si – •	109	109		
109	109	109		• – Si – •	110	110		
144	147	144	144	Si – O <sub>b</sub> – Si	131	124	145	139

Fig. 1. Calculated structures of  $\text{H}_3\text{Si-O-SiH}_3$  and  $(\text{OH})_3\text{Si-O-Si(OH)}_3$ , at the HF and B3LYP levels, compared with the experimental data [44,45] and higher level theoretical values [12,13] available. Bond lengths are in pm and bond angles in degrees. ● stands for terminal H atom or OH groups, respectively.

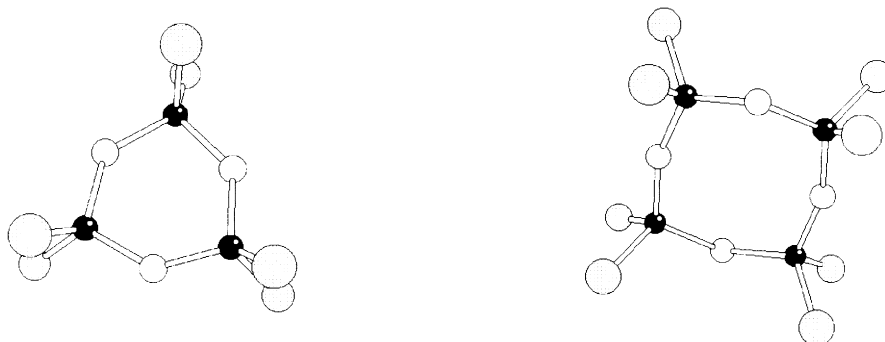
compared with the available experimental data and higher level calculation results. As it can be seen, there is a general agreement between the present results and the reported ones, at both levels of calculations, the B3LYP values being slightly better.

Present calculations and previous MP2/6-31G\* calculations [47] yield nearly identical geometrical parameters for the cyclosiloxanes (Fig. 2). The notable exception is the Si–O–Si bond angle of the tetramer form, with calculated values of 144



HF	B3LYP	MP2/6-31G*		HF	B3LYP	Exp.	MP2/6-31G*
163	165	166	Si – O	162	164	163	165
147	148	148	Si – H	147	148	148	148
108	108	108	O – Si – O	110	110	112	111
133	132	132	Si – O – Si	142	144	149	159

Fig. 2. Calculated structures of  $\text{Si}_3\text{O}_3\text{H}_6$  and  $\text{Si}_4\text{O}_4\text{H}_8$  at the HF and B3LYP levels, compared with the reported X-ray data for the tetramer form [46] and MP2/6-31G\* results [47] for both species (bond lengths are in pm and bond angles in degrees). Silicon, oxygen and hydrogen atoms are represented by black, white and grey circles, respectively.



HF	B3LYP	Exp.	MP2/6-31G*		HF	B3LYP	Exp.	MP2/6-31G*
161	164	162	165	Si – O	160	162	159	163
202	203	200	202	Si – Cl	201	203	200	202
110	110	110	111	Cl – Si – Cl	110	110	110	111
107	107	108	108	O – Si – O	110	110	110	110
133	133	132	132	Si – O – Si	160	160	149/171	160

Fig. 3. Calculated structures of  $\text{Si}_3\text{O}_3\text{Cl}_6$  and  $\text{Si}_4\text{O}_4\text{Cl}_8$  at the HF and B3LYP levels compared with the reported X-ray data and MP2/6-31G\* results [47] (bond lengths are in pm and bond angles in degrees). Silicon, oxygen and chloro atoms are represented by black, white and grey circles, respectively.

and  $159^\circ$ , respectively. The experimental value of  $149^\circ$  [46] obtained at the B3LYP level for the  $S_4$  crown-like configuration is found to be ca.  $1.4 \text{ kJ mol}^{-1}$  above the boat-like minimum shown in Fig. 2.

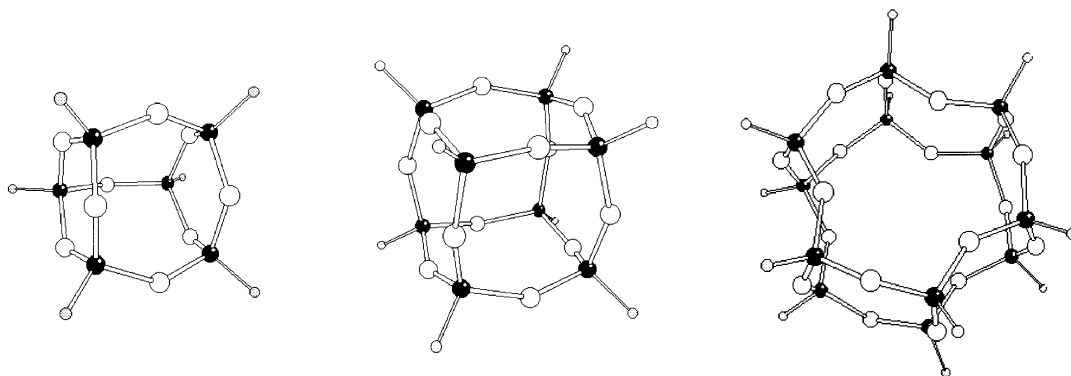
In the case of chlorosiloxanes (Fig. 3), the calculated structural parameters compare nicely with both the X-ray crystallographic data and the MP2/6-31G\* results, also reported in Ref. [47]. According to the X-ray results, the trimer presents an almost planar  $D_{3h}$  geometry in the crystal with three equivalent Si–O–Si bond angles, while the tetramer assumes a chair-like configuration, with two different types of Si–O–Si bond angles. The MP2/6-31G\* calculations yield planar structures for both forms, suggesting that the chair-like configuration of the tetramer arises from crystal packing constraints. As shown in Fig. 3, the calculated Si–O–Si bond angle for the planar trimer matches the experimental value of  $132^\circ$ , but the corresponding calculated value for the planar tetramer equals the average of the two experimental values ( $149$  and  $171^\circ$ ).

As expected, based on ring constraint effects, the Si–O–Si and O–Si–O bond angles are more

closed in the trimer than in the higher derivative. The difference is, however, more marked for the Si–O–Si angle, showing that the silicon atoms impose their preference towards regular tetrahedral co-ordination on the bridging oxygen atoms. The same effect was observed for the cyclic methylcyclosiloxane  $((\text{OSiMe}_2)_n)$ , with  $n = 3\text{--}6$  free molecules [48].

Fig. 4 shows the calculated structures of some spherosiloxanes with 6, 8 and 12 silicon atoms. Spherosiloxanes are a group of silicon containing systems that, during recent years, have been extensively used as model compounds within zeolite and silicate chemistry, in what concerns their catalytic and analytical applications. One of those interesting compounds is the highly symmetrical octahydrido-silasesquioxane molecule of  $O_h$  symmetry ( $\text{H}_8\text{Si}_8\text{O}_{12}$ , Fig. 4) for which several experimental results have been reported [49,51–54], as its  $\text{Si}_8\text{O}_{12}$  core is often found in zeolites.

The comparison of the calculated and the experimental results for  $\text{H}_8\text{Si}_8\text{O}_{12}$  clearly shows that the current ECP combination is adequate for accurate calculations on this type of systems, comparing



	HF	B3LYP	6-31G*	NLDA	HF	B3LYP	Exp.	6-31G*	NLDA	HF	B3LYP	6-31G*	NLDA
Si–O	162	165/164	164/163	168	161	164	162	163	168	161	164/163	163/162	167
O–Si–O	106/110	107/110	106/109	111	110	110	110	109	111	110/109	110/109	110/109	113
Si–O–Si	130/137	130/137	131/139	129/130	148	148	148	149	145	144/148	147/150	150/155	163/143

Fig. 4. Calculated structures of the oxo-spheroxanes  $H_6Si_6O_9$ ,  $H_8Si_8O_{12}$  and  $H_{12}Si_{12}O_{18}$  at the HF and B3LYP levels, compared with the gas-phase diffraction data reported for the  $H_8Si_8O_{12}$  derivative [49] and the 6-31G\* and NLDA values reported for the three systems [18,50–54] (bond lengths are in pm and bond angles in degrees). Silicon, oxygen and hydrogen atoms are represented by black, white and grey circles, respectively. For the  $H_6Si_6O_9$  and  $H_{12}Si_{12}O_{18}$  systems, due to the presence of ring with different sizes, the differentiated values are shown as 3R/4R and 4R/6R ring, respectively, in accordance with the notation used elsewhere [18,50].

with non-local density approximation (NLDA) [18] and HF/6-31G\* [50] approaches. The  $H_6Si_6O_9$  (of  $D_{3h}$  symmetry),  $H_8Si_8O_{12}$  and  $H_{12}Si_{12}O_{18}$  (of  $D_{6d}$  symmetry) systems (Fig. 4) follow the structural trends already observed for the previous systems. The Si–O–Si angle is particularly sensitive to the decrease of ring strain in the larger systems, while the remaining parameters are less sensitive. In fact, the Si–O–Si bond angle widens from  $130^\circ$  in the three-membered rings of  $H_6Si_6O_9$  to  $150^\circ$  in the four-membered rings of  $H_8Si_8O_{12}$  and  $155^\circ$  in the six-membered rings of  $H_{12}Si_{12}O_{18}$ , while the O–Si–O angle changes within the narrow 107–110° range.

#### 4. Conclusions

The practical application of the present ECP/basis set combination to the structural studies on both main group heptoxides and large polysilicate systems is strongly supported by the results herein reported.

The calculated lowest energy structures of the main group heptoxides are in good agreement

with the experimental data available, particularly in what concerns the relevant X–O–X bond angle parameter. The observed tendency to linearity of the X–O–X framework downwards a group and backwards a period can be explained by steric effects arising from the X-atom size and from the net charge of the heptoxide. The (p–d)-type interactions, found to be the main factor in determining the structure of transition metal heptoxides [1], do not play a significant role in the main group analogues.

In what concerns the polysilicate systems, particular attention was given to the Si–O–Si bond angle parameter, as the wide variety of the reported values reflect the ease in which this angle can vary to meet the structural requirements. Calculated Si–O–Si bond angles ranging from  $132^\circ$  in small rings to  $150^\circ$  in less strained systems have been found. As reported elsewhere [2], the decrease of the Si–O–Si bond angle is generally followed by a lengthening of the Si–O bonds, a behaviour that can be qualitatively related with the relative importance of the s- and p-orbitals in the oxygen atom hybridization. The results herein reported compare with the experimental



data available, and their accuracy compete with that of the results from much heavier 6-31G\* calculations.

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