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Free-volume studies in polyurethane membranes by positron annihilation spectroscopy

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Abstract

The free-volume parameters in various urethane/urea membranes obtained by varying the ratio of the structural constituents, polypropylene oxide and polybutadiene, were studied by positron lifetime and Doppler broadening measurements. On bi-soft segment membranes, a correlation was found between the composition of membranes, the normalized free volume, the radii of the holes and gas permeability. However, the correlation is not clear when PU data are also considered, indicating that other features must also play an important role in the permeation mechanism. © 2003 Elsevier Science Ltd. All rights reserved.

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1. Introduction

Polyurethanes, extensively used in the chemical industry, usually consist of hard segments (urethane/ urea groups and/or low molecular weight chain extenders) and one or more soft segments (e.g., polyether/ polyesters). Depending on the processing parameters and on the type of hard and soft segments these can be totally mixed or phase-separated to different degrees (Cooper and Tobolsky, 1966; Blackwell and Gardner, 1979; Fridman and Thomas, 1980). This structure versatility enables the production of materials with a wide range of properties. One important application of urethane/urea membranes is the exploitation of their selective permeability in industrial separations, namely gas permeation and pervaporation.

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The free-volume concept is extensively adopted in polymer science to explain many properties at microscopic structural level and to relate them with macroscopic properties. Positron annihilation lifetime spectroscopy is an important method for studying subnanometer size holes and for the determination of their size distribution and free-volume fractions (e.g. Jean, 1993; Dlubek et al., 2002). In a polymer matrix, thermalized positrons can form positronium (Ps) that will be localized in free-volume holes. The primary annihilation mechanism of the long-lived triplet state of positronium (ortho-positronium, o-Ps) is the pick-off annihilation with electrons of the host medium, with annihilation parameters dependent on the physicochemical properties. The lifetime and intensity of this component can be translated into hole size and density of holes. Recently, it has been shown that the Doppler broadening method may also give an important contribution to these studies (e.g. Dlubek et al., 2000). Indeed, the W-parameter, extensively applied in solidstate physics investigations (Flores et al., 2002) seems to provide specific sensitivity to the molecular bounds and ordering of the polymer chains and for this reason the

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Doppler spectroscopy gains increasing importance and acceptance in polymer studies as well (e.g., Semaan et al., 2001).

In this work, urea/urethane membranes obtained through the introduction of two soft segments were studied with the aims of establishing how the freevolume hole properties are influenced by the phase separation or mixing of the polymer components and correlate them with the known permeability features.

2. Experimental

The polymer membranes were synthesized from two basic units: polypropylene oxide-based prepolymer with three isocyanate terminal groups (PU), supplied by Portuguese Hoechst, and polybutadiene diol (PBDO) from Aldrich, containing 20–30 wt% vinyl, 10–25 wt% *cis*-1,4 and 50–60 wt% *trans*-1,4 isomers. Dibutyltin dilauryate (DBTDL), supplied by Aldrich, was used as catalyst. Toluene, from Merck, was used as solvent.

Cross-linked membranes were prepared through the reaction of the isocyanate groups of PU with the hydroxyl groups of PBDO catalyzed by DBTDL at 70–80°C. The ratio of PU to PBDO was varied to yield cross-linked polymers with different compositions.

Measurements of positron lifetime (LT) were carried out on PU/PBDO samples with up to 75% PBDO. The LT spectra were recorded using a fast–fast time spectrometer (with BaF₂ and Pilot-U scintillators), with a time resolution of 220 ps FWHM for ⁶⁰Co. The positron source, ca. 7×10^5 Bq of ²²Na was sandwiched between two identical specimens cut from 2 mm thick polymer sheets. The LT spectra, with about 1.7×10^6 integral counts, were recorded at (298±0.5) K and fitted by applying a version of POSITRONFIT (Kirkegaard and Eldrup, 1974). Appropriate source correction was made.

Doppler broadening (DB) measurements were also performed. The DB data were analyzed in terms of the conventional shape parameters S and W.

Gas permeation measurements were performed for all membranes with different PU/PBDO ratios.

3. Results and discussion

The LT spectra were deconvoluted into three components with lifetimes τ_i and intensities I_i . Subscripts i = 1-3 are ascribed to *p*-Ps, free e⁺ and *o*-Ps, respectively. The lifetime τ_3 and intensity I_3 are associated with pick-off annihilation of *o*-Ps in free volumes. According to the simple model of Tao and Eldrup (Tao, 1972; Eldrup et al., 1981) for an *o*-Ps confined in an infinitely deep spherical potential well, there is a correlation between the free-volume hole radius, R, and the long-lived lifetime, τ_3 , in ns

$$\tau_3 = 0.5 \left[1 - \frac{R}{R_0} + \frac{1}{2\pi} \sin\left(\frac{2\pi R}{R_0}\right) \right]^{-1},\tag{1}$$

where $R_0 = R + \Delta R$, and $\Delta R = 1.656$ Å (Nakanishi et al., 1987). In terms of this free-volume model, the fractional free volume, FFV, is determined by the product of I_3 by the cavity volume $V_f = (4/3)\pi R^3$ and by a material-dependent constant (Liu et al., 1993).

The inclusion of I_3 on the calculations of FFV from PALS has been recently questioned on the basis of results on temperature dependent studies on a given amorphous polymer (Maurer and Schmidt, 2000; Schmidt and Maurer, 2000). However, above T_g , where our measurements were performed, the applicability of the free-volume model seems to be quite acceptable.

In the present study, where polymers with different chemical composition are measured at a constant temperature above T_g and for which appropriate values of the material-dependent normalization parameter are unknown, it is expected that not only V_f but also I_3 have to be explicitly considered to obtain a reasonable estimate for the variation of total amount of freevolume from PALS results. Accordingly, considering that the PALS method should not be regarded as an absolute but rather a relative measurement technique, we calculate a relative FFV value from the free-volume model, normalized to the PU value, as a characteristic measure.

The values of R and the normalized FFV parameter, calculated for polymers with different amounts of polybutadiene diol (PBDO) are shown in Fig. 1. A continuous decrease is observed for these parameters with increasing PBDO content until about 50%; for higher PBDO content the values remain about constant.

These data are in agreement with previous thermal analysis studies (Zhao and Pinho, 1999) showing two glass transition temperatures (T_g) for membranes with PBDO content below 50% due to the coexistence of two phases corresponding to separate domains of the two soft segments (polybutadiene and polypropylene oxide). The two distinct T_g transition temperatures get closer to each other with increasing PBDO content and only one transition temperature is observed for PBDO content above 50%, indicating the formation of a homogeneous phase (full molecular mixing of the two soft segments). Optical microscopy has also shown the presence of these phases.

The dependence of CO_2 permeability with PBDO content is shown in Fig. 2b. The PU membrane (0% PBDO), which is not a bi-soft segment membrane, presents a low CO_2 permeability in contrast to the high values of its pore radius and FFV, as determined by LT. The increase up to 100% in the permeability of the membranes when a low percentage of PBDO is added is not mirrored in the LT parameters that even show a

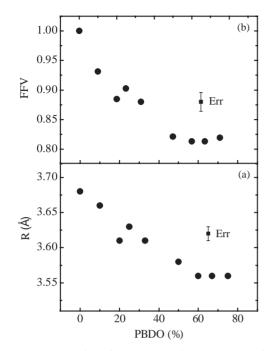


Fig. 1. Variation of the free-volume radius, R, (a) and of the relative fractional volume, FFV, normalized to the pure PU polymer (b), with increasing percentage of PBDO.

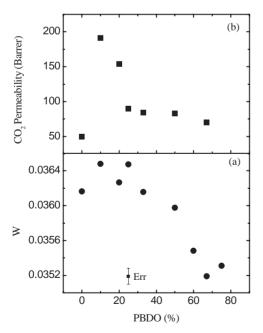


Fig. 2. Variation of the W-parameter (a) and the CO_2 permeation (b), with increasing percentage of PBDO.

decrease. These results strongly indicate that, besides the free volume probed by *o*-Ps, there are other factors involved in the permeation. Indeed, the interconnectivity

and transmissibility between holes may play a dominant role in that process. These features could be drastically changed by reordering with the formation of different free-volume patterns without a significant change in the overall amount of free volume. Comparing now only bisoft segment membranes, the highest permeability is obtained for low PBDO content. The increase in PBDO content favors the tendency for formation of more homogeneous membranes. The lower degree of segregation implies a decrease in the R and FFV values. This picture seems to be supported by DB measurements shown in Fig. 2a, where the W-parameter exhibits a maximum just in the concentration range where the gas permeability also has its maximum. This increase may be interpreted by assuming some ordering of polymer chains connected by chemical bounds where positrons can annihilate with high momentum electrons. This ordering could indeed provide an easier path for diffusion between holes. At this PBDO concentration range, the R and FFV values may even decrease.

Results of LT and DB measurements performed on some selected samples as a function of temperature, from room temperature until 50°C, show no change for samples in the homogeneous phase, whereas an increase of both R and FFV is found for the two-phase range. These data seem to support the picture described above, as in the homogeneous phase no ordering is expected.

In this picture, the fact that the highest values of R and FFV are obtained for the PU membranes, which present low gas permeation, may be considered as not really relevant since, at 0% PBDO content, we are not dealing with bi-soft segment membranes.

4. Conclusions

It is shown that hole size and overall free volume in bisoft segment polymer membranes, as obtained by positron annihilation, are correlated to gas permeation properties. However, the changes observed in those freevolume parameters are of much lower magnitude than the changes found in permeability and it can be concluded that hole size and overall free volume cannot by themselves describe those changes. The free volume seen by the *o*-Ps does not need to be identical to the free volume available for the gas permeation.

The extensive studies on gas permeation (e.g., Nagel et al., 2002) clearly indicate that the high complexity of the phenomena is influenced by many factors not discussed here due to space limitations, e.g., chain flexibility, chemistry at the free-volume walls, diffusion barriers, etc. As the gas permeation requests the actual physical passage of molecules across the polymer membrane, it seems to be also obvious to assume that the interconnectivity, which is not mirrored by the overall amount of the effective free volume seen by PALS must play a determining role in the outcome of the permeation features.

It is also not fully surprising that positrons, presenting only a static picture, cannot experience the driving forces due to concentration gradients and the size filtering across pathways through the wobbled interconnections among the holes whereas permeation is a dynamic process. A different diffusion mechanism may take place when the pattern changes from the pure PU to the bisoft system.

The presented work draws attention to the importance of such considerations, and also to the high additional value of Doppler measurements to the lifetime investigations, conventionally applied alone.

Further work along the above lines is in progress.

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